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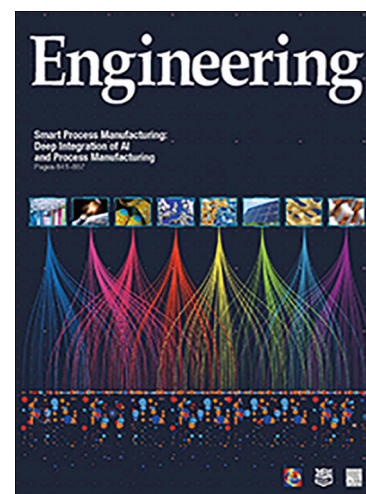
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Research

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# Robust Metal–Organic Frameworks with High Industrial Applicability in Efficient Recovery C<sub>3</sub>H<sub>8</sub> and C<sub>2</sub>H<sub>6</sub> from Natural Gas Upgrading

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**Abstract:** Developing efficient adsorbents with high uptake and selectivity for separation and recovery of C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> from natural gas is an important but challenging task. In this work, we demonstrate that high surface polarity and suitable pore diameter are two key factors that can synergistically enhance the separation performance, exemplified by metal–organic framework (MOF)-303 and MIL-160 (MIL: Matériaux de l’Institut Lavoisier), both possessing one-dimensional (1D) open channels with high density of heteroatoms and desired pore size (5–7 Å). Significantly, the uptake of MOF-303 for C<sub>3</sub>H<sub>8</sub> is up to 3.38 mmol·g<sup>−1</sup> at 298 K and 5 kPa with a record-high C<sub>3</sub>H<sub>8</sub>/CH<sub>4</sub> (5:85, v/v) ideal adsorbed solution theory (IAST) selectivity of 5114 among all reported MOFs. In addition, MOF-303 also displays high C<sub>2</sub>H<sub>6</sub> uptake capacity (at 10 kPa) and C<sub>2</sub>H<sub>6</sub>/CH<sub>4</sub> (10:85, v/v) selectivity, reaching 1.59 mmol·g<sup>−1</sup> and 26, respectively. Owing to the larger pore diameter and lower density of heteroatoms within its 1D channels, MIL-160 shows apparently lower uptake and selectivity compared to those of MOF-303, though the values exceed those of majority of reported MOFs. Density functional theory (DFT) calculations verify that the high surface polarity and the suitable pore diameter synergistically enhance the affinity of the frameworks toward C<sub>3</sub>H<sub>8</sub> and C<sub>2</sub>H<sub>6</sub>, giving rise to the high loading capacity and selectivity for C<sub>3</sub>H<sub>8</sub> and C<sub>2</sub>H<sub>6</sub>. Both MOFs feature remarkable moisture stability without structural change upon exposure to 95% relative humidity (RH) for a month. In addition, synthesis of both compounds can be readily scaled up through one-pot reactions to afford about 5 g samples with high crystallinity. Finally, the substantial potential of MOF-303 and MIL-160 as advanced adsorbents for efficient separation of C<sub>3</sub>H<sub>8</sub>/C<sub>2</sub>H<sub>6</sub>/CH<sub>4</sub> has been demonstrated by ternary breakthrough experiments, regeneration tests, and cyclic evaluation. The excellent separation performance, high stability, low cost, and good scalability endow both MOFs promising adsorbents for natural gas purification and recovery of C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub>.

**Keywords:** Metal–organic framework; Hydrocarbon adsorption and separation; Selectivity; Stability; Scale-up synthesis

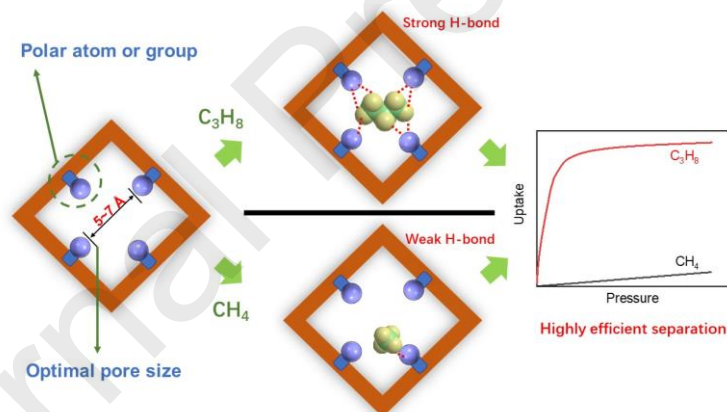
## 1. Introduction

As the result of rapid urbanization, industrialization, and human population growth, the consumption of fossil fuels has been increasing constantly and swiftly. Natural gas is one of the most commonly used energy sources not only due to its abundant reserves in nature but also because it is more eco-friendly compared to oil and coal [1,2]. Generally, besides CH<sub>4</sub> as the main component, C<sub>2</sub>H<sub>6</sub> (0–20%) and C<sub>3</sub>H<sub>8</sub> (0–5%) coexist in the natural gas and should be fully recovered individually for updating the quality of natural gas. Subsequently, the recovered C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> can be utilized to produce C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>6</sub> through catalytic dehydrogenation, which are the raw materials for producing various plastics, such as polyethylene, poly(vinyl chloride), and polypropylene [3–5]. Traditional cryogenic distillation has shown excellent performance in industrial separation processes but it is operated under high pressure and/or low temperature, leading to high energy-consumption [6,7]. Among the newly developed separation methods, adsorptive separation technology has been regarded as a promising alternative to cryogenic distillation because of its high separation efficiency and low energy input [8,9]. The applicability of adsorptive separation is largely dependent on the performance of adsorbents, which involves their porosity, pore dimensions, and pore surface functionality.

Metal–organic frameworks (MOFs) represent an emerging class of porous materials assembled from metal centers and organic ligands through coordination bonds, which have drawn tremendous attention in the field of gas separation, due to their intrinsic advantages including extra-high surface area, structural diversity, pore tunability, and controllable functionality [10–14]. While MOFs have shown encouraging potential in the separation and recovery of C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> from natural gas, there still exist several problems that need to be addressed. First, the uptake capacity of C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> at low pressure is still not sufficient. For example, boron cage pillared supramolecular metal–organic framework (BSF)-2 possesses a very high selectivity but the equilibrium uptake amounts of C<sub>3</sub>H<sub>8</sub> and C<sub>2</sub>H<sub>6</sub> are only 1.77 and 1.22 mmol·g<sup>−1</sup>, respectively, leading to low separation efficiency [15]. MIL-100(Fe) (MIL: Matériaux de l’Institut Lavoisier) exhibits high uptake capacity of C<sub>2</sub>H<sub>6</sub>

and  $C_3H_8$  at 1 bar (1 bar =  $10^5$  Pa) but the values are too low at low pressure [16]. As a result, these MOFs show limited applicability due to the low  $C_2H_6$  and  $C_3H_8$  concentrations in nature gas. Low stability of most reported MOFs in the presence of moisture is the second problem. For example, MOF-74 and HKUST-1 (HKUST: Hong Kong University of Science and Technology) feature high separation capability but suffer from structural decomposition in humid air [17,18]. High moisture stability is essential for an adsorbent to be used in nature gas purification because the gas mixture always contains a certain amount of water vapor [19–22]. In addition, material cost and scale-up capability are also important factors in determining the feasibility for industrial applications. While some MOFs show high performance and sufficient water/moisture stability, such as Ni(TMBDC)(DABCO) $_{0.5}$  (TMBDC: 2,3,5,6-tetramethylterephthalic acid; DABCO: 1,4-diazabicyclo[2.2.2]octane) [23] and FJI-C4 (FJI: Fujian Institute of Research on the Structure of Matter) [7], they are constructed from expensive organic linkers which would seriously hinder their industrial utility when considering the economic feasibility. MOFs synthesized in relative harsh conditions involving high temperature, high pressure, large amounts of template, or complex steps are usually difficult to scale up [24,25]. Hence, developing or screening ideal MOF adsorbents for targeted industrial applications require consideration of all these aspects in order to reach an optimal balance between separation performance, stability, and cost.

To improve the uptake of  $C_2H_6$  and  $C_3H_8$  at low pressure, we propose an effective strategy based on our previous study: simultaneously enhancing the polarity of the pore surface and tuning the pore diameter to a suitable size (Fig. 1). On one hand, increasing surface polarity by introducing polar atoms or functional groups could markedly strengthen the interaction between gas molecules and MOFs as the C–H bond in hydrocarbons preferentially binds to the polar atoms or functional groups through hydrogen bonds [11,26,27]. On the other hand, a suitable pore size is very crucial because the diffusion of  $C_3H_8$  molecules would be seriously restricted if pore is too narrow, and on the contrary, a larger pore always causes weaker interaction, leading to unsatisfactory uptake capacity and selectivity, especially at low pressure. We could envision that suitable pore diameter and highly polar surface simultaneously existing in a MOF could concurrently reinforce multiple hydrogen bonds, resulting in significant enhancement in high adsorption capacity and selectivity. As to moisture stability, MOFs built on metals with high valence such as  $Zr^{4+}$ ,  $Cr^{3+}$ , and  $Al^{3+}$  are generally resistant to water [28–32]. Finally, using MOFs with inexpensive ligands and high scale-up ability can lower the total cost and significantly enhance their likelihood for industrial applications.



**Fig. 1.** Schematic illustration on how highly polar surface and suitable pore diameter of a MOF may synergistically enhance the affinity of the framework toward  $C_3H_8$  to result in a highly efficient separation of  $C_3H_8$  and  $CH_4$ .

MOF-303 and MIL-160 are selected in this work when taking all these factors into account. Their ultrahigh hydrothermal stability, large surface area, and especially the suitable pore diameter, as well as the high density of heteroatoms in the one-dimensional (1D) channel synergistically afford an intriguing performance in natural gas purification and recovery of  $C_2H_6$  and  $C_3H_8$ . Adsorption isotherms for  $CH_4$ ,  $C_2H_6$ , and  $C_3H_8$  were determined by a volumetric method. The ideal adsorbed solution theory (IAST) selectivity of  $C_2H_6/CH_4$ ,  $C_3H_8/CH_4$ , and  $C_3H_8/C_2H_6$  at various ratios, as well as the isosteric heats of adsorption for all three gases were calculated. Ternary fixed-bed breakthrough experiments were carried out to further confirm their potential for real-world applications. Finally, density functional theory (DFT) calculation was applied to study the adsorption mechanism for  $CH_4$ ,  $C_2H_6$ , and  $C_3H_8$ .

## 2. Materials and methods

### 2.1. Reagents and solvents

All reagents were purchased commercially and used as received. Aluminum chloride hexahydrate was purchased from Alfa Aesar (USA); 3,5-pyrazoledicarboxylic acid monohydrate and 2,5-furandicarboxylic acid were both supplied by TCI

America (USA); sodium hydroxide was purchased from Acros Organics (USA); the high-purity gases for adsorption experiments were obtained from Praxair, Inc. (USA).

## 2.2. Preparation of MOF-303 and MIL-160

MOF-303 was prepared by using the procedure reported by Yaghi et al. [33] with some modifications. First, 1.04 g aluminum chloride hexahydrate ( $\text{AlCl}_3 \cdot 6\text{H}_2\text{O}$ , 4.308 mmol) and 0.75 g 3,5-pyrazoledicarboxylic acid monohydrate ( $\text{H}_3\text{PDC}$ , 4.308 mmol) were dissolved in 72 mL water in a 200 mL glass flask, 3 mL aqueous sodium hydroxide ( $\text{NaOH}$ , 0.26 g, 6.5 mmol) were added dropwise to the above mixture under stirring. The flask was then heated at 100 °C with reflux for 12 h. After cooling down to room temperature, the as-synthesized MOF-303 powder was obtained by filtration. To remove the remaining 3,5-pyrazoledicarboxylic acid, the powder was washed thoroughly with water, followed by heated under vacuum at 150 °C for 12 h. MIL-160 was synthesized by the same procedure as that of MOF-303 by replacing the ligand 3,5-pyrazoledicarboxylic acid monohydrate with 2,5-furandicarboxylic acid.

## 2.3. Characterization

Powder X-ray diffraction (PXRD) patterns were collected on a Bruker D8 Venture X-ray diffractometer (Bruker, USA) with a  $2\theta$  range of 3°–35° at  $2.0^\circ \cdot \text{min}^{-1}$ . Thermogravimetric (TG) measurements were performed on a TA Q-5000 apparatus (TA Instruments, USA), with a ramping rate of  $10 \text{ K} \cdot \text{min}^{-1}$  from ambient temperature to 973 K under a flowing nitrogen environment. Nitrogen ( $\text{N}_2$ ) adsorption isotherms were obtained on a Micromeritics 3Flex analyzer (Micromeritics Instrument Corporation, USA) at 77 K. The Brunauer–Emmett–Teller (BET) model was chosen to evaluate the specific surface area, and the Horvath–Kawazoe (HK) method was conducted to acquire the micropore size distribution.

## 2.4. Adsorption experiments

$\text{CH}_4$ ,  $\text{C}_2\text{H}_6$ , and  $\text{C}_3\text{H}_8$  adsorption isotherms were performed on the 3Flex analyzer. Volumetric sorption data were collected at varied temperatures and pressures up to 1 bar. Approximately 80–100 mg of sample were used and degassed at 423 K for 12 h prior to the adsorption experiments.

## 2.5. Breakthrough experiments

Breakthrough curves of a ternary mixture  $\text{CH}_4/\text{C}_2\text{H}_6/\text{C}_3\text{H}_8$  (85:10:5, v/v/v) were obtained on a homemade experimental setup under the control of a mass flow meter, with the flow rate set to be  $2 \text{ mL} \cdot \text{min}^{-1}$ . A small adsorption column was prepared by packing about 0.2 g of activated sample into a long stainless hollow cylinder. The real time concentration of the effluent component was monitored by gas chromatography (Agilent, USA). Before the experiment, the packed column was heated at 423 K under He flow ( $5 \text{ mL} \cdot \text{min}^{-1}$ ) for 1 h. After the breakthrough experiment, the desorption curves were collected at 323 K under  $5 \text{ mL} \cdot \text{min}^{-1}$   $\text{N}_2$  flow.

## 2.6. Theoretical calculations

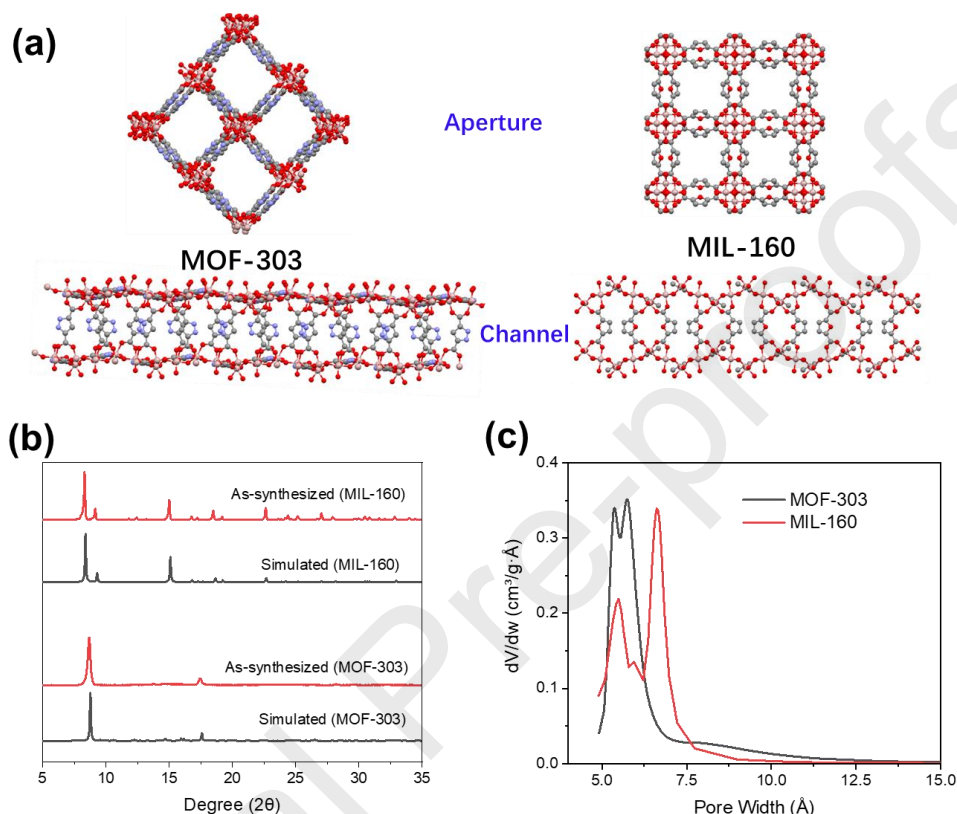
All *ab initio* calculations were performed using DFT in Vienna *ab initio* simulation package (VASP) [34,35], with van der Waals density functional (vdW-DF) method [36–39] to take into account of important van der Waals interactions. All the MOF unit cells were optimized by carrying out spin-polarized calculations, with self-consistent field (SCF) convergence of 0.1 meV ( $1 \text{ meV} = 1.6 \times 10^{-22} \text{ J}$ ) and the plane wave energy cut-off set at 600 eV. The unit cell parameters and ions were allowed to move till the force acting between atoms reached below  $5 \text{ meV} \cdot \text{\AA}^{-1}$ . Potential binding sites were studied by placing  $\text{CH}_4$ ,  $\text{C}_2\text{H}_6$ , and  $\text{C}_3\text{H}_8$  molecules in MOF-303 and MIL-160 at various sites and all the atoms were allowed to relax in accordance with the convergence condition. Difference in the total energies of the MOF unit cells and the guest molecules was used to calculate the corresponding binding energies. Induced charge densities were also calculated that maps the variation in charge density upon introduction of the guest molecules and help identifying the interactions happening at the binding sites.

# 3. Results and discussion

## 3.1. Material characterization

MOF-303 and MIL-160 are built from 1D infinite chain of  $[\text{Al}(\text{OH})(\text{COO})_2]_n$  linked through 3,5-pyrazoledicarboxylate and 2,5-furandicarboxylate, respectively, to form three-dimensional (3D) networks with straight 1D open channels, as shown in Fig. 2(a). Phase purity of the powder samples was confirmed by PXRD analysis (Fig. 2(b)). The PXRD patterns of as-synthesized samples match well with the corresponding simulated patterns.  $\text{N}_2$  sorption experiments were conducted at 77 K to establish permanent microporosity of the samples. As shown in Figs. S1 and S2 (in Appendix A), the  $\text{N}_2$  uptake of MOF-

303 and MIL-160 increases sharply and reaches saturation at very low pressure with high adsorption capacities, indicating high microporosity of the two MOFs. MOF-303 and MIL-160 exhibit large BET surface area of 1220 and 1188  $\text{m}^2\cdot\text{g}^{-1}$ , respectively, and uniform micropore with size of 5–7 Å (Fig. 2(c)), which are well-suited for adsorption of  $\text{C}_3\text{H}_8$  and  $\text{C}_2\text{H}_6$  at low concentrations. Thermogravimetric analysis (TGA) was carried out to investigate the thermal stability of the two MOFs. As shown in Fig. S3 in Appendix A, both MOF-303 and MIL-160 show two distinct weight loss steps: The first step before 125 °C corresponds to the physically adsorbed solvent molecules, and the second step following the plateau starting from 420 and 350 °C relates to structure decomposition, indicating their high thermal stability.

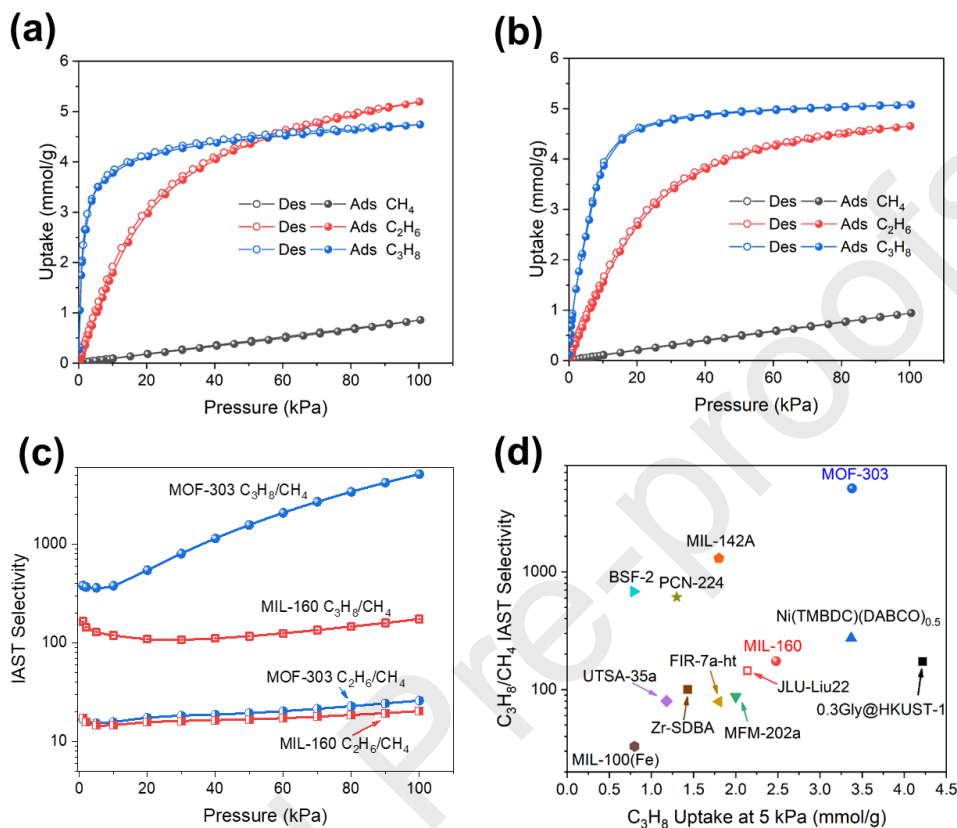


**Fig. 2.** (a) Pore aperture and 1D channel of MOF-303 and MIL-160. Color scheme: red, grey, pink, and light blue balls represent O, C, Al, and N atoms, respectively. (b) Experimental and simulated PXRD patterns of MOF-303 and MIL-160. (c) Pore size distribution of MOF-303 and MIL-160.

### 3.2. Single-component adsorption isotherms of $\text{CH}_4$ , $\text{C}_2\text{H}_6$ , and $\text{C}_3\text{H}_8$

The high stability, large surface area, and especially the suitable pore diameter, as well as the high density of N or O atoms decorating the 1D channels of the two MOFs prompted us to investigate their separation performance for natural gas purification/upgrading and recovering of  $\text{C}_3\text{H}_8$  and  $\text{C}_2\text{H}_6$  from the gas mixture. Single-component adsorption isotherms of  $\text{CH}_4$ ,  $\text{C}_2\text{H}_6$ , and  $\text{C}_3\text{H}_8$  for MOF-303 and MIL-160 were collected at 298 K (Figs. 3(a) and (b)). Both  $\text{C}_3\text{H}_8$  isotherms of MOF-303 and MIL-160 exhibit type I adsorption profile with very steep slopes at low pressure and reach saturation below 20 kPa, indicating strong interaction between  $\text{C}_3\text{H}_8$  molecule and the frameworks. Notably, at 298 K and 1 bar, MOF-303 and MIL-160 adsorb 4.74 and 5.08  $\text{mmol}\cdot\text{g}^{-1}$  of  $\text{C}_3\text{H}_8$ , respectively. At a low pressure of 5 kPa relevant to the typical partial pressure of  $\text{C}_3\text{H}_8$  in nature gas, the corresponding uptakes remain as high as 3.38 and 2.48  $\text{mmol}\cdot\text{g}^{-1}$ , much higher than those of many reported MOFs, such as UTSA-35a (UTSA: University of Texas at San Antonio) [40], FIR-7a-ht (FIR: Fujian Institute of Research) [41], and MIL-100(Fe) [16], but lower than Mg/Fe/Co-MOF-74 (about 4.1  $\text{mmol}\cdot\text{g}^{-1}$ ) [17], 0.3Gly@HKUST-1 (4.22  $\text{mmol}\cdot\text{g}^{-1}$ ), and crystalline porous material (CPM)-734c (about 3.7  $\text{mmol}\cdot\text{g}^{-1}$ ) [42]. This indicates the potential of MOF-303 and MIL-160 for the recovery of low pressure  $\text{C}_3\text{H}_8$  from  $\text{CH}_4$  stream. As shown in Figs. 3(a) and (b), the adsorption isotherms of  $\text{C}_2\text{H}_6$  can also be characterized as type I but with much smaller slope compared to those of  $\text{C}_3\text{H}_8$ , which may be ascribed to the weaker interaction between  $\text{C}_2\text{H}_6$  molecule and the frameworks. The  $\text{C}_2\text{H}_6$  uptakes are as high as 4.96 and 4.65  $\text{mmol}\cdot\text{g}^{-1}$  at 298 K and 1 bar for MOF-303 and MIL-160, respectively. At a relatively low pressure (10 kPa) the values are 1.59 and 1.55  $\text{mmol}\cdot\text{g}^{-1}$ , respectively, outperforming many reported MOFs, such as ZnSDB ( $\text{H}_2\text{SDB}$ : 4,4'-sulfonyldibenzoic acid) [43], MIL-142A [44], and porous coordination network (PCN)-224 [45], except Mg/Fe/Co-MOF-74

(about  $3.2 \text{ mmol} \cdot \text{g}^{-1}$ ) [17,46],  $\text{Co}_2\text{V-bdc-tpt}$  (bdc: terephthalate; tpt: 2,4,6-tri(4-pyridyl)-1,3,5-triazine; about  $2.3 \text{ mmol} \cdot \text{g}^{-1}$ ) [47], and  $\text{Ni}(\text{TMBDC})(\text{DABCO})_{0.5}$  (about  $2.93 \text{ mmol} \cdot \text{g}^{-1}$ ) [23]. This further confirms that the two MOFs are promising candidates for simultaneous capture of  $\text{C}_3\text{H}_8$  and  $\text{C}_2\text{H}_6$  from nature gas. As for  $\text{CH}_4$ , on the contrary, the isotherms are nearly straight lines with relatively low equilibrium uptake capacities of  $0.86$  and  $0.94 \text{ mmol} \cdot \text{g}^{-1}$  for MOF-303 and MIL-160, respectively, suggesting weak affinity of the frameworks toward the  $\text{CH}_4$  molecules.



**Fig. 3.** Adsorption (Ads)–desorption (Des) isotherms of  $\text{C}_3\text{H}_8$ ,  $\text{C}_2\text{H}_6$ , and  $\text{CH}_4$  on (a) MOF-303 and (b) MIL-160 at 298 K. (c) IAST selectivities of MOF-303 and MIL-160 for  $\text{C}_3\text{H}_8/\text{CH}_4$  (5:85, v/v) and  $\text{C}_2\text{H}_6/\text{CH}_4$  (10:85, v/v) at 298 K. (d) Comparison of IAST selectivities and adsorbed amounts of  $\text{C}_3\text{H}_8$  at 298 K and 5 kPa on MOF-303, MIL-160, and previously reported top-performing MOFs (JLU: Jilin University; MFM: Manchester framework material; H<sub>2</sub>SDBA: 4,4'-sulfonfylidibenzoic acid).

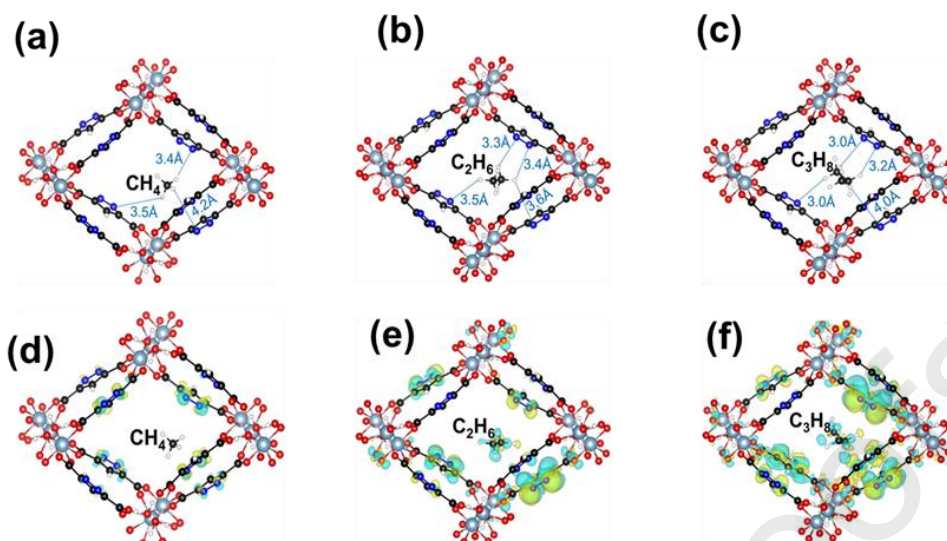
To quantify the extent of interactions between the frameworks and the three hydrocarbons, isosteric heats of adsorption ( $Q_{\text{st}}$ ) for  $\text{CH}_4$ ,  $\text{C}_2\text{H}_6$ , and  $\text{C}_3\text{H}_8$  were calculated (Figs. S6 and S7 in Appendix A). The zero coverage  $Q_{\text{st}}$  for MOF-303 follows the following sequence:  $\text{C}_3\text{H}_8$  ( $34 \text{ kJ} \cdot \text{mol}^{-1}$ ) >  $\text{C}_2\text{H}_6$  ( $24 \text{ kJ} \cdot \text{mol}^{-1}$ ) >  $\text{CH}_4$  ( $19 \text{ kJ} \cdot \text{mol}^{-1}$ ), which is consistent with the order of the adsorbed amounts. Similarly, for MIL-160, the  $Q_{\text{st}}$  values follow the same order:  $\text{C}_3\text{H}_8$  ( $35 \text{ kJ} \cdot \text{mol}^{-1}$ ) >  $\text{C}_2\text{H}_6$  ( $28 \text{ kJ} \cdot \text{mol}^{-1}$ ) >  $\text{CH}_4$  ( $19 \text{ kJ} \cdot \text{mol}^{-1}$ ). The apparent differences for the three gas species may result from their different polarizability and molecule size. For  $\text{C}_3\text{H}_8$ , having the highest polarizability ( $6.3 \times 10^{-24} \text{ cm}^3$ ) among the three and the largest molecule dimension that is close to the pore size of the two MOFs guarantee that appreciably strong hydrogen bonds could be formed with the frameworks upon its adsorption. In contrast,  $\text{CH}_4$  possesses a relatively low polarizability ( $2.6 \times 10^{-24} \text{ cm}^3$ ) and a much smaller molecule size, leading to insufficient contacts with the pore surface and a much weaker interaction. The distinct differences in  $Q_{\text{st}}$  and adsorption capacity for  $\text{CH}_4$ ,  $\text{C}_2\text{H}_6$ , and  $\text{C}_3\text{H}_8$  make MOF-303 and MIL-160 excellent adsorbents for recovering  $\text{C}_3\text{H}_8$  and  $\text{C}_2\text{H}_6$  from natural gas. To more precisely evaluate the separation performance of the two MOFs under mixed-gas conditions, we calculated IAST selectivities of  $\text{C}_3\text{H}_8/\text{CH}_4$  (5:85, v/v) and  $\text{C}_2\text{H}_6/\text{CH}_4$  (10:85, v/v) at room temperature (298 K). As shown in Fig. 3(c), for MOF-303, the selectivity of  $\text{C}_3\text{H}_8/\text{CH}_4$  (5:85, v/v) at 100 kPa reaches 5114, which, to the best of our knowledge, has set a new record. In addition, the selectivity for  $\text{C}_2\text{H}_6/\text{CH}_4$  (10:85, v/v) is as high as 26, surpassing many reported MOFs (Fig. S7). Compared to MOF-303, MIL-160 possesses a similar  $\text{C}_2\text{H}_6/\text{CH}_4$  (10:85, v/v) selectivity (20) but a much lower  $\text{C}_3\text{H}_8/\text{CH}_4$  (5:85, v/v) selectivity (174), mainly attributed to the lower  $\text{C}_3\text{H}_8$  uptake amount at low pressure and the slightly larger adsorbed amount of  $\text{CH}_4$ .

As a comparison, the uptake capacities of  $\text{C}_3\text{H}_8$  and  $\text{C}_2\text{H}_6$  combined with the  $\text{C}_3\text{H}_8/\text{CH}_4$  and  $\text{C}_2\text{H}_6/\text{CH}_4$  selectivities for MOF-303, MIL-160, and some representative MOFs are listed in Fig. 3(d) and Fig. S8 (in Appendix A). The  $\text{C}_3\text{H}_8/\text{CH}_4$  selectivity of MOF-303 (5114) is the highest value reported by far and notably higher than those of top-performing MOFs,

including Ni(TMBDC)(DABCO)<sub>0.5</sub>, BSF-2, and MIL-142A. Additionally, the C<sub>3</sub>H<sub>8</sub> uptake amount (3.38 mmol·g<sup>-1</sup>) is comparable to Ni(TMBDC)(DABCO)<sub>0.5</sub> (3.37 mmol·g<sup>-1</sup>) and exceeds other MOFs, such as UTSA-35a, Manchester framework material (MFM)-202a [48], Zr-SDBA (H<sub>2</sub>SDBA: 4,4'-sulfonyldibenzoic acid) [49], and MIL-142A. For C<sub>2</sub>H<sub>6</sub>/CH<sub>4</sub>, as shown in Fig. S7, MOF-303 also outperforms most of the other MOFs with higher selectivity (26) and uptake capacity (1.59 mmol·g<sup>-1</sup>), such as BSF-2, ZnSDB, and UTSA-35a, but lower than that of Ni(TMBDC)(DABCO)<sub>0.5</sub> and Mg/Co/Fe-MOF-74 [17,46]. The ultrahigh selectivity combined with the large adsorption amount of C<sub>3</sub>H<sub>8</sub> and C<sub>2</sub>H<sub>6</sub> at both 100 kPa and low pressure (5 or 10 kPa) suggests that MOF-303 is of great promise for use as adsorbent in natural gas upgrading. Both the uptake amounts and selectivities of MIL-160 are slightly lower than those of MOF-303, which may be attributed to the larger pore size and the lower density of heteroatoms in MIL-160. Regardless, MIL-160 still features higher uptake and selectivity than the majority of the MOFs listed. It should be noted that while it seems MgMOF-74 and Ni(TMBDC)(DABCO)<sub>0.5</sub> are more promising for C<sub>2</sub>H<sub>6</sub> recovery (Fig. S8) in terms of adsorption capacity and selectivity, the former suffer from low moisture resistance, and the latter is composed of an expensive ligand (TMBDC). In contrast, both MOF-303 and MIL-160 feature high thermal and moisture stability. To further confirm their outstanding hydrothermal stability, PXRD patterns of samples were collected after being treated under different conditions, including exposure to moisture for one month. As presented in Figs. S9–S12 (in Appendix A), compared to the results of as-synthesized samples, there are no notable differences found in the PXRD patterns of the samples after various treatments, confirming the excellent hydrothermal stability of both materials. As for the cost, the ligands of MOF-303 and MIL-160 are both inexpensive, and the MOFs could be readily synthesized under mild conditions. Furthermore, to assess the scale-up capability, we synthesized the two MOF sample in 5 g scale. High-quality powder products were obtained by constant stirring the mixture of ligands, aluminum salts, water, and NaOH in a glass flask at 100 °C for 12 h, with the yield higher than 90% (Figs. S13–S18 in Appendix A). The high moisture stability, low cost, and the facile scale-up capability further corroborate the feasibility for the implementation of MOF-303 and MIL-160 in industrial separation technology.

### 3.3. Density functional theory analysis

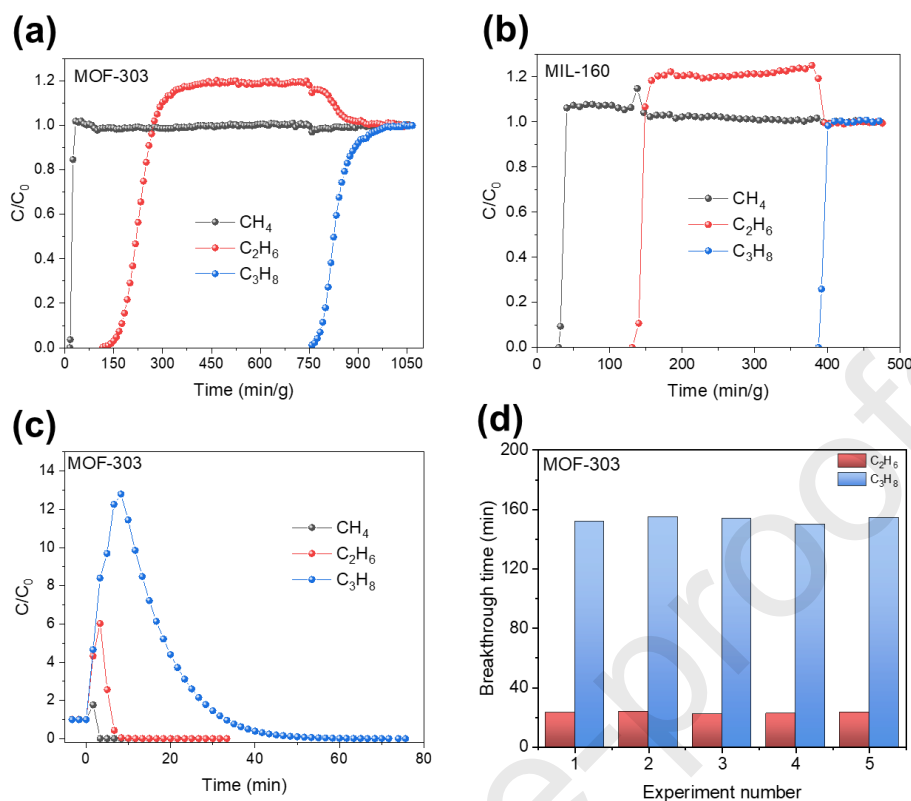
To understand the gas adsorption behavior and mechanism in MOF-303 and MIL-160, DFT calculations were performed on both structures. As shown in Figs. 4(a)–(c), all three guest molecules at their primary binding positions occupy similar sites in the pore of MOF-303. The binding energy calculations show a binding strength order of C<sub>3</sub>H<sub>8</sub> (60.1 kJ·mol<sup>-1</sup>) > C<sub>2</sub>H<sub>6</sub> (47.4 kJ·mol<sup>-1</sup>) > CH<sub>4</sub> (30.7 kJ·mol<sup>-1</sup>) at their primary binding sites located near the MOF linker. This result agrees with the experimental uptake trends for MOF-303 at low pressures. The induced charge density for the three gas molecules (Figs. 4(d)–(f)) shows most charge rearrangement occurring between the guest molecules and the N atoms in the linker, confirming that the N atoms are the strongest binding sites in MOF-303. As shown in Figs. 4(c) and (f), the highest binding energy and induced charge density for C<sub>3</sub>H<sub>8</sub> are likely a result of its longer C3 chain, larger kinetic diameter, and a greater number of C–H arms as compared with C<sub>2</sub>H<sub>6</sub> and CH<sub>4</sub>, which enable it to interact more closely with multiple linkers and form multiple stronger hydrogen bonds. The charge rearrangements for MIL-160 are shown in Fig. S19 (in Appendix A). All three guest molecules prefer binding near the linkers and interact primarily with the O atoms on both the furan ring and the carboxyl groups via their C–H bonds. Different from their binding sites in MOF-303, CH<sub>4</sub> and C<sub>2</sub>H<sub>6</sub> molecules bind close to linkers on one side of pore (Figs. S19(d)–(e)), which may be attributed to the larger pore size of MIL-160 (Fig. 2(c)). On the other hand, C<sub>3</sub>H<sub>8</sub>, owing to its larger size, occupies a more central position in the pore and interacts with linkers on three different sides. The binding energy calculations for MIL-160 generate the same trend as that of MOF-303, with the descending order of C<sub>3</sub>H<sub>8</sub> (65.7 kJ·mol<sup>-1</sup>) > C<sub>2</sub>H<sub>6</sub> (55.8 kJ·mol<sup>-1</sup>) > CH<sub>4</sub> (34.9 kJ·mol<sup>-1</sup>) at the primary binding site (Figs. S19(a)–(c)), in consistent with the experimental observations.



**Fig. 4.** The unit cell of MOF-303 used for calculating binding energies. The results for (a) CH<sub>4</sub>, (b) C<sub>2</sub>H<sub>6</sub>, and (c) C<sub>3</sub>H<sub>8</sub> show their primary binding site (color scheme: silver, red, black, blue, and white balls represent Al, O, C, N, and H). Induced charge densities for (d) CH<sub>4</sub>, (e) C<sub>2</sub>H<sub>6</sub>, and (f) C<sub>3</sub>H<sub>8</sub> (iso-level = 0.001 electrons·Å<sup>-3</sup>). Blue highlighted areas represent a depletion of charge and yellow areas represent an increase in charge after the guest molecule occupies a binding site.

### 3.4. Breakthrough experiments, adsorbent regeneration, and recyclability tests

To further evaluate the actual separation potential of MOF-303 and MIL-160, dynamic breakthrough experiments were carried out on ternary (CH<sub>4</sub>/C<sub>2</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>8</sub>, 85:10:5, v/v/v) gas mixtures at 298 K. For the breakthrough curves of MOF-303, as shown in Fig. 5(a), CH<sub>4</sub> eluted out first at 15 min·g<sup>-1</sup>, followed by C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> with a breakthrough time of 120 and 760 min·g<sup>-1</sup>, respectively, coinciding well with the order of static capacities and isosteric heats. The distinct breakthrough times for the three gases confirm that MOF-303 can afford efficient natural gas upgrading by the individual C<sub>3</sub>H<sub>8</sub> and C<sub>2</sub>H<sub>6</sub> recovery (as shown in Fig. S20 in Appendix A). The breakthrough curves of MIL-160 are similar to those of MOF-303 but exhibit a much shorter breakthrough time for C<sub>3</sub>H<sub>8</sub> (388 min·g<sup>-1</sup>), as shown in Fig. 5(b), which is consistent with the lower static adsorption capacity of MIL-160 for C<sub>3</sub>H<sub>8</sub> at 5 kPa. To evaluate the regeneration property of MOF-303 and MIL-160, desorption was carried out under dynamic N<sub>2</sub> flowing at 333 K. As shown in Fig. 5(c), during the desorption process for MOF-303, CH<sub>4</sub>, C<sub>2</sub>H<sub>6</sub>, and C<sub>3</sub>H<sub>8</sub> can be fully removed in 4, 12, and 50 min, respectively, demonstrating that MOF-303 column can be fully regenerated under mild condition. Finally, five breakthrough measurements were carried out consecutively to evaluate the recycling performance of MOF-303 and MIL-160. As illustrated in Fig. 5(d) and Fig. S21 (in Appendix A), the breakthrough times for C<sub>3</sub>H<sub>8</sub> and C<sub>2</sub>H<sub>6</sub> are nearly constant, suggesting good recyclability of both adsorbents.



**Fig. 5.** Ternary breakthrough curves ( $CH_4/C_2H_6/C_3H_8$ , 85:10:5, v/v/v) of (a) MOF-303 and (b) MIL-160 at 298 K. (c) Desorption curves of MOF-303 packed column at 333 K under  $5 \text{ mL} \cdot \text{min}^{-1}$   $N_2$  flow. (d)  $C_3H_8$  and  $C_2H_6$  breakthrough times of MOF-303 in consecutive five cycles of breakthrough experiments (298 K and 100 kPa).

#### 4. Conclusion

In summary, two highly stable Al-based MOFs, MOF-303 and MIL-160 were investigated for efficient separation and recovery of  $C_2H_6$  and  $C_3H_8$  from natural gas. The large surface area, suitable pore diameter, and the high density of N or O atoms in the 1D open channels synergistically enhance the affinity of the frameworks toward  $C_3H_8$  and  $C_2H_6$ , resulting in large uptake capacity and excellent selectivity. In particular, for MOF-303, the uptake of  $C_3H_8$  at 298 K and 5 kPa is up to  $3.38 \text{ mmol} \cdot \text{g}^{-1}$  along with a record-high IAST selectivity of 5114 for  $C_3H_8/CH_4$  (5:85, v/v). MOF-303 also possesses high adsorption capacity for  $C_2H_6$  (at 10 kPa) and  $C_2H_6/CH_4$  (10:85, v/v) selectivity, reaching as high as  $1.59 \text{ mmol} \cdot \text{g}^{-1}$  and 26, respectively. DFT calculations verified that the strong affinity between  $C_3H_8$  (or  $C_2H_6$ ) molecules and the heteroatoms on the linkers gives rise to the high loading capacity and selectivity for  $C_3H_8$  and  $C_2H_6$ . To investigate the potential of using MOF-303 and MIL-160 to upgrade natural gas and to recover  $C_3H_8$  and  $C_2H_6$ , further experiments, including hydrothermal stability and scale-up capability tests, as well as breakthrough experiments, were carried out on both MOFs. The stability tests revealed their exceptional resistance to moisture. The scale-up capabilities were verified through a synthesis at 5 g scale with quantitative yields. Finally, ternary breakthrough experiments and recyclability tests further confirmed the great potential of MOF-303 and MIL-160 as advanced adsorbents for efficient separation of  $C_3H_8/C_2H_6/CH_4$ .

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#### Compliance with ethics guidelines

Shikai Xian, Junjie Peng, Haardik Pandey, Timo Thonhauser, Hao Wang, and Jing Li declare that they have no conflict of interest or financial conflicts to disclose.

## Appendix A. Supplementary data

Supplementary data to this article can be found online at

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